## **Supplementary Materials**

### **A deep neural network potential model for transition metal diborides**

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#### **Supplementary Text 1. Crystal structure of transition metal diborides (TMB2)**

Transition metal diborides (TMB<sub>2</sub>) crystalize to AlB<sub>2</sub>-like structure with the space group of P6/mmm, stacking alternatively by close-packed TM layers and graphite-like hexagonal B layers, as illustrated in Supplementary Figure 1. TM atoms and boron atoms are respectively located at 1a (0, 0, 0) and 2d (1/3, 2/3, 1/2) Wyckoff positions.



**Supplementary Figure 1.** Schematic illustration on the crystal structure of transition metal diboride. The big spheres are transition metals, while the small spheres represent boron.



**Supplementary Figure 2.** Phonon dispersion curves of TMB<sub>2</sub>s.

**Supplementary Table 1. Melting points (in K) measured by experiments (exp),calculated by the DP model (cal) or averaged from constitute mono-compounds (ave).Experimental data are from ref [1]**



#### **Supplementary Text 2. Method on the construction of grain boundaries**

Symmetric tilt grain boundaries are constructed based on coincident site lattice (CSL) theory, which are similar to our previous work [2]. When exact CSL cannot be defined, small elastic strains should be applied to the system. The tilt axes are [010] and [210] of diboride, respectively. Supplementary Figure 3 takes the 010 21 grain boundary as an example to illustrate how a grain boundary is defined. A grain boundary plane can be defined by two in-plane vectors, which are contained in its name.One is the rotation axis of the symmetric tilt grain boundary. The other is a combination of other two basis vectors. For example, the 010 21 grain boundary has a rotation axis of [010] and another in-plane direction of  $2[001]+[2\overline{1}0]$ , i.e., [2 $\overline{1}2$ ], while the 210–12 grain boundary has a rotation axis of [210] and another in-plane direction of [001]+2[010]. Supplementary Table 1 lists the crystallographic features of the fourteen grain boundaries.



**Supplementary Figure 3.** Illustration on the definition of 010\_21 grain boundary. Red points represent one diboride lattice comprising the grain boundary, while blue points represent the other diboride lattice. Green points defines the CSL points.

**Supplementary Table 2. Crystallographic information of symmetric tilt grain boundaries studied in this work. The four-index representation of the vectors are also listed for better understanding**



[010] and [210] are tilt axis.

Basis vectors of [010] tilt grain boundaries are [001] and [ $2\overline{1}0$ ] for one crystal, [00 $\overline{1}$ ] and [ $2\overline{1}0$ ] for the other crystal; while basis vectors of [210] tilt grain boundaries are [001] and [010] for one crystal,  $[00\overline{1}]$  and  $[010]$  for the other crystal.

The headlines means linear combination of basis vectors to define CSL vectors, e.g., 21 of [010] tilt grain boundary means a CSL vector in the grain boundary plane is linear combination of two [001] basis vector and one [ $\overline{21}0$ ] basis vector, i.e., [ $\overline{21}2$ ].



**Supplementary Figure 4.** Relaxed atomic structures for the grain boundaries studied in this work.<br>010\_22, and 010\_33 are the same as 010\_11 grain boundary, while 210\_22, and 210\_33 are the same as 210 11 grain boundary. In the 210 21 grain boundary, a circle indicates the presence of local free volume, potentially accommodating additional atoms. Green atoms are boron, and blue atoms are metals. When drawing B-B bonds, the criterion is set to be 2 Å.

**Supplementary Table 3. Ideal strength (in GPa) of grain boundaries. Without\_MC means the metallic sites are randomly occupied, while with\_MC means the last snapshot after MC/MD simulation is adopted for tensile simulation. Without\_W means no W is added, while with\_W means 5 at% W is added into metallicsites**



During tensile simulation, stress variation is approximately 1 GPa in the tensile direction. Consequently, when the strength difference is less than 1 GPa, distinguishing which grain boundary is stronger becomes inconclusive due to the margin of error.

# **Supplementary** Text 3. The predictive capabilities of the DP model on grain boundary energies **and grain boundary segregation energies.**

In this section, we illustrate the predictive capabilities of the DP model on grain boundaries by examining the grain boundary energies and grain boundary segregation energies, and comparing the results to DFT calculations. Owing to the computational limitations of DFT, analyses are constrained to small supercells. We chose the 010-21 and 210-11 grain boundaries for this study, depicted in Supplementary Figure 5A and B, respectively. Notably, the 010\_21 structure here deviates from that shown in Supplementary Figure 4. To ascertain the grain boundary energy accurately, it is requisite to construct a supercell that mirrors the bulk phase's chemical composition, maintaining a 1:2 metal-to-boron ratio. Consequently, we refrain from removing boron atoms adjacent to the grain boundary (highlighted by an ellipse), a decision that leads to pronounced distortion of the boron framework, as evident from Supplementary Figure 5A.

The calculated grain boundary energies of  $TMB_2$  compounds (TM = Ti, Zr, Hf, V, Nb, Ta, Cr, Mo, W) using both the DP model and DFT are shown for comparison in Supplementary Table 4. The accuracy for TiB<sub>2</sub>,  $ZrB_2$ , and HfB<sub>2</sub> is found to be high, moderate for  $VB_2$ ,  $NbB_2$ , and  $TaB_2$ , and low for CrB2, MoB2, and WB2. However, the trend of grain boundary energy variation with the transition metal is correctly captured. In  $TiB<sub>2</sub>$ ,  $ZrB<sub>2</sub>$ , and  $HfB<sub>2</sub>$ , the grain boundaries possess high energy; in VB2, NbB2, and TaB2, medium energy; and in CrB2, MoB2, and WB2, low energy. Notably, the DFT-computed grain boundary energies for WB<sub>2</sub> are negative, highlighting its inherent instability. Moreover, very low grain boundary energies are also observed for CrB<sub>2</sub> and MoB<sub>2</sub>, suggesting a predisposition of Cr, Mo, and W to segregate at the grain boundaries due to favorable local atomic structures.

Segregation energies were also computed for various solutes (Ti, Zr, Hf, V, Nb, Ta, Cr, Mo, W) within  $TiB<sub>2</sub>$ ,  $ZrB<sub>2</sub>$ , and  $HfB<sub>2</sub>$  grain boundaries. Solute insertion sites are marked by arrows in Supplementary Figure 5A and B, with the supercells featuring solute atoms at grain boundary sites (highlighted by a blue arrow) as the reference state. The difference in energy between this state and the reference state defines segregation energy. Supplementary Figure 5C presents acomparison of segregation energies derived from both the DP model and DFT calculations, demonstrating a good correspondence.

When generating the training data for the DP model, no grain boundary structure is included. Nevertheless, as shown by the grain boundary energies and segregation energies, the predictive capabilities of the DP model on grain boundaries are reasonable.

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**Supplementary Table 4. Comparison of grain boundary energies (meV/Å<sup>2</sup>) calculated by the DP model and DFT calculations**

		TiB <sub>2</sub>							$ZrB_2$   HfB <sub>2</sub>   VB <sub>2</sub>   NbB <sub>2</sub>   TaB <sub>2</sub>   CrB <sub>2</sub>   MoB <sub>2</sub>   WB <sub>2</sub>	
010 21	<b>DFT</b>	256	174	191	188	112	91	82	10	$-52$
	DP	207	141	156	88	134	123	100	84	33
210 11	<b>DFT</b>	140	142	143	67	54	34	13		$-29$
	DP	156	151	155	115	103	88	62	38	$\mathfrak b$



**Supplementary Figure 5.** Atomic structures of (A) the 010\_21 grain boundary and (B) the 210\_11 grain boundary constructed for the assessment of grain boundary energy and segregation energy; (C) Segregation energy (*ΔE*) comparisons between DFT calculations and the DP model. The matrix for blue, orange and green points is TiB<sub>2</sub>, ZrB<sub>2</sub>, HfB<sub>2</sub>, respectively.

#### **References**

[1] Rudy E, Harmon DP. Ternary phase equilibria in transition metal-boron-carbon-silicon systems. Part V. compendium of phase diagram data. Tech. Rep. AFML-TR-65-2, Part V, Air Force Materials Lab., Air Force Systems Command, Wright-Patterson Air Force Base, OH, 1969: 5.

[2] Dai FZ, Wen B, Xiang H, Zhou Y. Grain boundary strengthening in  $ZrB_2$  by segregation of W: atomistic simulations with deep learning potential. *J Eur Ceram Soc* 2020;40:5029-36. [DOI: 10.1016/j.jeurceramsoc.2020.06.007]